Crumpling wires in two dimensions

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An energy-minimal simulation is proposed to study the patterns and mechanical properties of elastically crumpled wires in two dimensions. Our aim is to describe the behavior at the intermediate occupancy of the cavity so that its radius of gyration is varied up to one twentieth of the wire length. We tuned the bending rigidity and stretching modulus to measure the energy allocation, size-mass exponent, and the stiffness exponent. The mass exponent is shown to be constant at value D_M =1.33, so is the stiffness exponent α =−0.25. But the latter varies with the plasticity parameters *s* and θ_p . These numerical findings agree excellently with the experimental results.

DOI: [10.1103/PhysRevE.78.067101](http://dx.doi.org/10.1103/PhysRevE.78.067101)

PACS number(s): $62.20 - x$, $05.90 + m$, 89.75 .Da

I. INTRODUCTION

The process of crumpling is everywhere in nature and human activities, including the formation of mountains and valleys in tectonics $[1]$ $[1]$ $[1]$, packing of DNA strands in viruses [[2](#page-3-1)], car wreckage after an accident $\lceil 3 \rceil$ $\lceil 3 \rceil$ $\lceil 3 \rceil$, the noisy food wraps that irritate us in the theater, etc. In spite of its ubiquity, the mechanism behind many of its properties has remained unclear $[4,5]$ $[4,5]$ $[4,5]$ $[4,5]$. For instance, how does the labyrinthian internal structure evolve to withstand extraordinary pressure while more than 80% of its interior remains vacant? Also, why is there a power law between the external force and the sphere radius with an exponent that varies with material $\lceil 6 \rceil$ $\lceil 6 \rceil$ $\lceil 6 \rceil$ but, otherwise, is insensitive to the thickness and size of the thin sheet?

Scientists have studied crumpled wires (CWs) both theoretically $\lceil 7-9 \rceil$ $\lceil 7-9 \rceil$ $\lceil 7-9 \rceil$ and experimentally $\lceil 10-12 \rceil$ $\lceil 10-12 \rceil$ $\lceil 10-12 \rceil$. As is shown in Fig. [1,](#page-0-0) the wire is smooth and rid of the complicated ridges and vertices, which is different from a crumpled sheet $\lceil 13 \rceil$ $\lceil 13 \rceil$ $\lceil 13 \rceil$. In the last few years, more and more interesting properties of CWs have emerged. For instance, Donato *et al.* found a scal-ing law in the size-mass relation [[10](#page-3-8)], while Stoop *et al.* offered the associated morphological phase diagram and reported a power law for the number of loops with different exponents in each morphology $[7]$ $[7]$ $[7]$. However, there are still few quantitative studies of CWs either on the energy aspect or macroscopic properties. Furthermore, the former authors all worked on stuffing the wire into a fixed two-dimensional cavity. It is not clear whether this shares the same properties as by increasing the strength of the confining potential while fixing the wire length. Aside from this, another motivation is to check how sensitive the CW is on the specific form and relative amplitude of stretching and bending energies. In this report, simulations are carried out to study the mechanical properties of CWs under the semiflexible polymer $\lceil 9 \rceil$ $\lceil 9 \rceil$ $\lceil 9 \rceil$ and the minimal energy models $[14]$ $[14]$ $[14]$.

II. MODEL AND COMPUTATIONAL METHOD

In our simulations, the optimal shape of a wire is obtained by minimizing the total energy of CWs. We begin with a wire in a weak external field, and gradually increase the field strength. At each stage with a certain field strength, we recursively minimize the energy and move to the next stage with a larger external field after an equilibrium state has been achieved. This process is carried out throughout the whole simulation until the wire structure no longer changes with the increasing external field. The wire consists of *N* number of monomers and is of length $L_0=1$. The total energy of a CW includes bending energy E_b , stretching energy E_s , and external potential *E*ext. In addition, a hardcore potential *E*hc is incorporated to keep the wire from self-crossing. Explicitly, the energy function can be written as

$$
E_{\text{tot}} = \sum_{i=1}^{N-1} \frac{k_b}{2} \theta_i^2 + \sum_{i=1}^{N} \left[\frac{k_s}{2} (\Delta l_i)^2 + \lambda r_i^2 \right] + E_{\text{hc}},\tag{1}
$$

where θ_i is the bending angle between two successive segments and $k_b = \frac{YI}{(L_0/N)}$ is the bending rigidity which is proportional to the Young's modulus *Y* and second moment of inertia *I*. The stretching modulus $k_s = YA/(L_0/N)$ depends on the wire cross section *A*, and Δl_i is the deviation of the *i*th segment from its equilibrium length L_0/N . The coefficient λ

FIG. 1. A crumpled copper wire in a two-dimensional cavity. This photo is reproduced by courtesy of Donato *et al.* [[10](#page-3-8)].

TABLE I. The value of parameters used in simulations.

Parameter	Range of values
Bending rigidity k_h	1 to 10
Stretching modulus k_{s}	10^4 to 10^7
External field λ	1 to 2×10^3
Number of segments N	100
Wire length L_0	
Width of hard core	10^{-5}

of the quadratic potential can be controlled to simulate different strength of crumpling force, and the distance r_i of the *i*th monomer is measured from the center of the external parabolic potential. Since most research interests fall into cases with insignificant extension, we mainly focused on conditions that $k_b \ll k_s$. Typical parameters for simulations are listed in Table [I.](#page-1-0) The steady state of CW is determined by minimizing its total energy with Powell's algorithm. To make sure that the local minimum we found was not a special case, we adopted a random set of searching directions in Powell's algorithm and repeated the simulations for ten times to get the averaged results. Most of the time, CW are randomly folded rather than forming a perfect spiral—the apparent global minimum $\lceil 16 \rceil$ $\lceil 16 \rceil$ $\lceil 16 \rceil$. This indicates that those folded configurations are metastable states. The modified Powell's minimization is equivalent to the Monte Carlo simulation in this aspect.

III. RESULTS AND DISCUSSION

A. Shape and size-mass exponent

The loops of CW in Figs. [1](#page-0-0) and [2](#page-1-1) resemble the boundary of a water-drop-like structure. Compare with the simulations for a semiflexible polymer $[9]$ $[9]$ $[9]$ which explicitly put in finite temperatures and consequently resulting in many small fluctuations, the configuration of CW is characteristically smooth, and so we exclude the temperature effect in this simulation. We believe the smoothness is credited to the stretching force which causes the relaxation of roughness, analogous to the surface tension which disfavors a kinky surface. Because the stretching energy is found to be insignificant when $k_b \ll k_s$, we can neglect it when determining the mathematical form of the configuration. For simplicity and without loss of generality, we consider the case of $N \rightarrow \infty$,

i.e., the continuous limit. Equation ([1](#page-0-1)) then can be rewritten as

$$
E_{\text{tot}}^* = \int_0^L \left[\frac{k_b}{2} \left(\frac{d\theta}{ds} \right)^2 + \lambda r^2 \right] ds.
$$
 (2)

The search for shape function $r(\theta)$ with minimal energy is done by introducing the variational method. Fix both $r(s)$ $r(s=L)$ and this total energy is identical to the classical action once we identify the arc length parameter *s* as the time. The Euler-Lagrange equation of motion immediately gives out the shape function as $r^2 \sqrt{k_b/\lambda} = \sin 2\theta$, where the r^2 term is not hard to expect by the dimensional analysis in retrospect. This matches the pattern in the initial stage of the crumpling observed experimentally $\lceil 10 \rceil$ $\lceil 10 \rceil$ $\lceil 10 \rceil$ as well as in our simulations. However, when the loops start to touch each other as the crumpling proceeds further, the excluded volume or steric interaction $\lceil 15 \rceil$ $\lceil 15 \rceil$ $\lceil 15 \rceil$ is expected to squeeze and distort the shape of loops.

The mass function $M(R)$ is a common property $[10]$ $[10]$ $[10]$ to characterize the configuration of a CW. It is defined as the total mass encompassed within a circular area of radius *R* centering at the CW's center-of-mass:

$$
M(R) = \sum_{i=1}^{N-1} \theta(R - r_i) \delta[\theta(r_i - R), \theta(r_{i+1} - R)] \frac{1}{N}
$$

+ $\delta[\theta(r_i - R), \theta(R - r_{i+1})] \frac{1}{N} f(r_i, r_{i+1}),$ (3)

where $\theta(x)$ denotes the step function, $\delta(x, y)$ stands for the Kronecker δ function which equals 1 when $x=y$ and zero otherwise, and $f(r_i, r_{i+1})$ is the portion of length between r_i and r_{i+1} within the circle. After performing a series of simulations, the mass-size relation is computed and shown in Fig. [3](#page-2-0) for six distinct combinations of k_b and k_s . We can see that the mass of the CW grows in a power-law fashion when the radius increases and, disregarding $\lceil 10 \rceil$ $\lceil 10 \rceil$ $\lceil 10 \rceil$ the fluctuations at small external field, the growing exponent saturates at D_M $=1.33$ for $\lambda > 100$; see Fig. [3.](#page-2-0) The saturating behavior is in accordance with the conjecture of Donato *et al.* [[10](#page-3-8)] that in a loose-packing situation the corresponding mass-size exponent has insignificant dependence on how the wires are injected. And the value of D_M =1.33 is also consistent with the result of Donato *et al.*

FIG. 2. A sequence of simulation results for $k_b = 5$ and $k_s = 5 \times 10^5$. The value of λ for (a), (b), (c), and (d) are 70, 200, 300, and 1000, respectively. It can be seen that when we increase λ , no sharp edge occurs and the shape of CW remains smooth. Those local structures start to contact each other when $\lambda \ge 1000$. More turning points are formed if the external field increases.

FIG. 3. The mass-size relation of CW for $\lambda = 1000$. The radius of the circle is rescaled by R_G . Apparently, all mass will already be included in the circle when $R=1$. The rescaling process only shifts the data on a log-log plot and does not affect the value of exponent D_M = 1.33. Since we set the origin of the circle at the center-of-mass of the CW, there is a bias in the data at small *R*. This bias is avoided by fitting only from $R=0.5$ to $R=1$ as shown by the solid straight line $\lceil 10 \rceil$ $\lceil 10 \rceil$ $\lceil 10 \rceil$.

B. Energy allocation

We characterize the allocation of elastic energies in the CW by the ratio $\gamma = E_s/E_b$. Data in Fig. [4](#page-2-1) establish a scaling relation between γ and the three parameters in the Hamiltonian as

$$
\gamma \sim \lambda^{\epsilon_l} k_b^{\epsilon_b} k_s^{\epsilon_s},\tag{4}
$$

where ϵ_l =0.55, ϵ_b =2.6, and ϵ_s =−3.3. The numerical results can be deduced analytically. These exponents can be obtained by doing a dimensional analysis on the bending en-ergy and the external potential in Eq. ([2](#page-1-2)), $k_b / l \sim \lambda l^3$, which gives the characteristic size for each loop as $l \sim (k_b/\lambda)^{1/4}$. Multiplying the bending energy stored in each loop k_b/l by the number of loops *L*/*l* enable us to estimate the bending

FIG. 4. The value of γ is plotted against several parameters. For (a) and (b), γ grows with the strength of external field in a powerlaw fashion with exponent ϵ_l = 0.55. In (a) the stretching modulus is set at 2×10^3 and in (b) the bending rigidity is 6. Panels (c) and (d) depict γ versus k_b and k_s , respectively, and they share the same denotation of data. Notice that the power law can only be found for strong external field ($\lambda \ge 500$). The exponents of the power law are determined to be ϵ_b =2.6 and ϵ_s =−3.3.

FIG. 5. The shrinkage of R_G versus λ for CW with different rigidity and modulus, k_b and k_s , with the same notation as in Fig. [3.](#page-2-0) All data collapse to the parameters $k_b=8$, $k_s=2.4\times10^5$ with the scaling $R_g \sim (k_b k_s)^{1.2}$. The solid line is the fitting result $R_g \sim \lambda^{-0.25}$. The inset depicts the number of loops N_l versus $1/R_G$ for which the slope is measured to be 1.6, consistent with the value 1.8 obtained by Ref. $[7]$ $[7]$ $[7]$.

energy as $k_b L/l^2$. In a similar fashion, the stretching energy can be written as $E_s \sim k_s (L - L_0)^2$. Minimizing the sum of these two energies with respect to *L* gives

$$
\gamma = \frac{\sqrt{k_b \lambda}}{k_s [L_0 - \sqrt{k_b \lambda}/k_s]}.
$$
\n(5)

The value of ϵ _l=0.5 matches the simulation result. Although the exact magnitudes are not the same, the signs of ϵ_b and ϵ_s and their relative magnitude are still captured. One possible explanation for the discrepancy is our failure to include the hardcore potential in the analytic arguments, which has been known to be difficult $\lceil 15 \rceil$ $\lceil 15 \rceil$ $\lceil 15 \rceil$ but crucial for a realistic polymer.

C. Stiffness and plasticity

How to estimate the resistance of a crumpled structure is a well-known difficult problem. For instance, what is the main reason why such a loose crumpled structure can be so hard? People have studied the power law between the structure size and the applied force both experimentally and theoretically $[6,17]$ $[6,17]$ $[6,17]$ $[6,17]$. Our simulation result in Fig. [5](#page-2-2) showed that the radius of gyration $R_g = r_{\text{max}}$ decreases with λ in a powerlaw form. To prevent the structure from becoming too complicated, we limit λ at 2000 to optimize the resolution. The data are found to collapse into the scaling relation R_g $\sim \lambda^{\alpha} (k_b k_s)^{1.2}$, which suggests the stretching energy not only correlates those monomers, but also contributes to the stiffness. Intuitively, the dependence of k_s is expected to drop out instead of diverging as it approaches infinity. The reason why we obtained such a scaling relation is a consequence of the discrete model since the length of monomers is bound to be compressed as they all scramble to be near the center of the external potential.

In Fig. [5,](#page-2-2) $\alpha = -0.25$ is determined to be independent of k_b and *ks* and, consequently, the thickness and Young's modulus of the wire. This result is the same as the predictions made for an elastic thin sheet $\lceil 18 \rceil$ $\lceil 18 \rceil$ $\lceil 18 \rceil$. It can be derived analytically by minimizing with respect to *R* the sum of the external poten-

tial $\lambda R^2 L_0$ and the total repulsive interaction $\beta (L_0 / R^2)^2 R^2$, where β is a measure of the hardcore potential, L_0 / R^2 is the monomer density, and the other R^2 factor comes from integrating over the cavity. In order to establish a link with the previous approach, some analysis on the morphology is done. In the inset of Fig. [5,](#page-2-2) the number of loops $N_l \equiv \sum_i |\pi|$ $-\theta_i / \pi$ is plotted as a function of the inverse of R_G . These two quantities are observed to obey the same power-law relation as obtained by Ref. $[7]$ $[7]$ $[7]$ with a similar exponent.

Since Stoop *et al.* have demonstrated the importance of plasticity for CW, we also include it in our simulation. The bending rigidity k_b is revised with a linear approximation $sk_b(0 \le s \le 1)$ in the plastic regime, so that the bending energy becomes $E_b = k_b \theta_p^2 / 2 + sk_b (\theta^2 - \theta_p^2) / 2 + k_b \theta_p (1 - s)(\theta - \theta_p)$ when θ is larger than the yield threshold angle θ_p [[7](#page-3-6)]. We analyzed the $\lambda - R_g$ relation of a CW for $s = 0.05$, 0.1, 0.2, 0.4, and 0.8, and it behaves very differently from that of the elastic one. In Fig. [6,](#page-3-16) the value of α decays with the bending rigidity in the plastic regime which is consistent to the experimental findings in Ref. $[6]$ $[6]$ $[6]$. It also decays exponentially with the yield threshold θ_p . When large deformations cost less energy due to the plasticity, they appear in abundance and cause many vertex-like structures which makes the CW stiffer.

IV. CONCLUSION

In conclusion, we introduced a minimal one-dimensional model to simulate crumpled wires. The simulation results agree excellently with the experimental observation, which include the pattern of loops and the constant value of masssize exponent D_M =1.33 [[10](#page-3-8)]. We checked the exponent α

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FIG. 6. The $R_g - \lambda$ relation for CW $[(k_b, k_s) = (5.5 \times 10^5)]$ with different s as denoted in the figure. The value of exponents α are −0.21, −0.24, and −0.29 for *s*=0.4, 0.2, and 0.05. The inset illustrates that α decays with θ_p exponentially and saturates at value 0.26 for *s*=0.4.

that characterizes the power-law relation between the crumpling force and the CW radius to be independent of the bending rigidity and the stretching modulus which is consistent with Ref. [[18](#page-3-15)]. Plasticity is found to suppress α which is in the right trend as the recent experiment $\lceil 6 \rceil$ $\lceil 6 \rceil$ $\lceil 6 \rceil$ on crumpled sheets.

ACKNOWLEDGMENTS

We benefited from correspondence and discussions with Professors P. C. Chen, H. J. Herrmann, P. Y. Hsiao, Y. Kantor, H. H. Lin, and E. Terentjev. Support by the National Science Council in Taiwan is acknowledged.

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